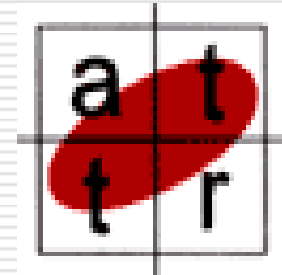
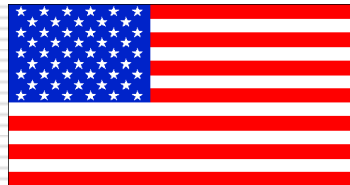


Enhanced Quality in Electric Melt Grey Cast Irons

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OUTLINE

□ INTRODUCTION

- * **Grey Iron: 48% of World Metalcasting Production**
- * **Important Changes in Grey Iron Melting Technologies**

□ HETEROGENEOUS GRAPHITE NUCLEATION IN GREY IRONS

- * **Three – Stage Model of Graphite Nucleation in Grey Iron**

□ LOW S / Al, SUPERHEATED GREY CAST IRONS

- * **High Eutectic Undercooling and Chill Tendency**

□ METALLURGICAL TREATMENT OF ELECTRIC MELT IRONS

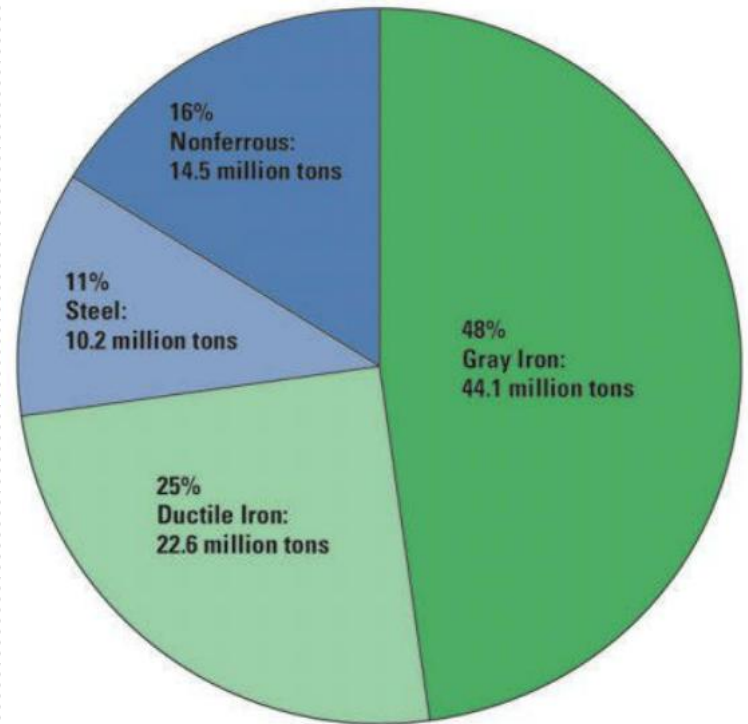
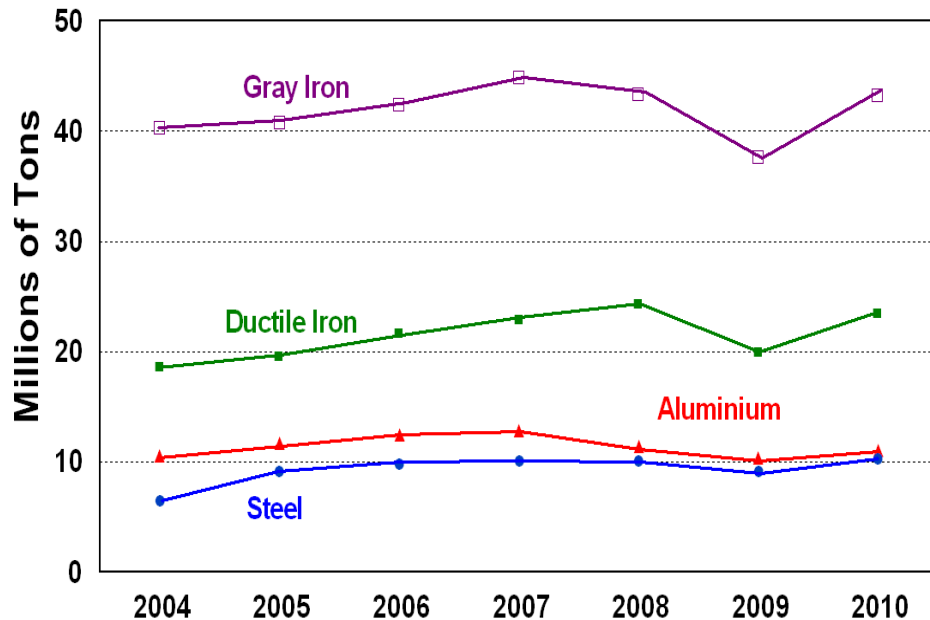
- * **Pre-treatment of the Iron Melt in a Melting Furnace**
 - Carbon Material / Metallurgical Silicon Carbide / Aluminium
- * **Double Treatments [preconditioning + inoculation]**
 - **Al** preconditioning + **Ca-FeSi** inoculation
 - **Zr** preconditioning + **Sr-FeSi** inoculation
 - **Al,Zr,Ca-FeSi** preconditioning + **Ca,Ba-FeSi** inoculation

□ RECOMMENDATIONS FOR FOUNDRY APPLICATION

FOUNDRIY METALLIC MATERIALS EVOLUTION WORLDWIDE

Grey Iron – The largest contributing material to global production

2010 World Casting Production



[Modern Casting – Census of World Casting Production, 2004 – 2010]

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❖ **IMPORTANT CHANGES IN GREY IRONS PRODUCTION**

➤ **Historically: Cupola Melted Irons**

* **$> 0.5\% \text{Mn}$, $> 0.05\% \text{S}$, $> 0.005\% \text{Al}$, $(\% \text{Mn}) \times (\% \text{S}) > 0.03$**

* favourable conditions for MnS – type compounds

- major nucleation sites for flake graphite / low chill tendency

➤ **Small or/and inefficient Cupolas – replaced by a new generation of Coreless Induction Furnaces**

[200 – 1000 Hz, $> 250 \text{ kW/t}$, acid lined]

* High melting rate, no heel, steel charge, stirring, $> 1500^\circ\text{C}$

* **$> 0.5\% \text{Mn}$, $< 0.05\% \text{S}$, $< 0.005\% \text{Al}$, $(\% \text{Mn}) \times (\% \text{S}) < 0.03$**

- high chill / undercooled graphite tendency

➤ **Grey, Ductile & Compacted Graphite Irons**

Production in the same Foundry

* Advantage : a single Low S Base Iron, single Inoculant, $> 1500^\circ\text{C}$

* **$< 0.5\% \text{Mn}$, $< 0.03\% \text{S}$, $< 0.005\% \text{Al}$, $(\% \text{Mn}) \times (\% \text{S}) < 0.02$**

- the highest chill / undercooled graphite

[M. Chisamera, S. Stan, I. Riposan G. Costache, M. Barstow – 112 AFS Casting Congr., 2008, Atlanta, USA]

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The main objective of the present paper

- ❖ **To explore what conditions are needed to improve the quality of electric melt grey iron**
 - superheated in acid lined, medium frequency coreless induction furnaces,
 - especially for thin wall iron castings production [automotive industry]

- ❖ **A review of original data obtained by the authors,** *relating to*
 - heterogeneous graphite nucleation model in commercial grey irons, *and*
 - specific solidification pattern of low S - irons,

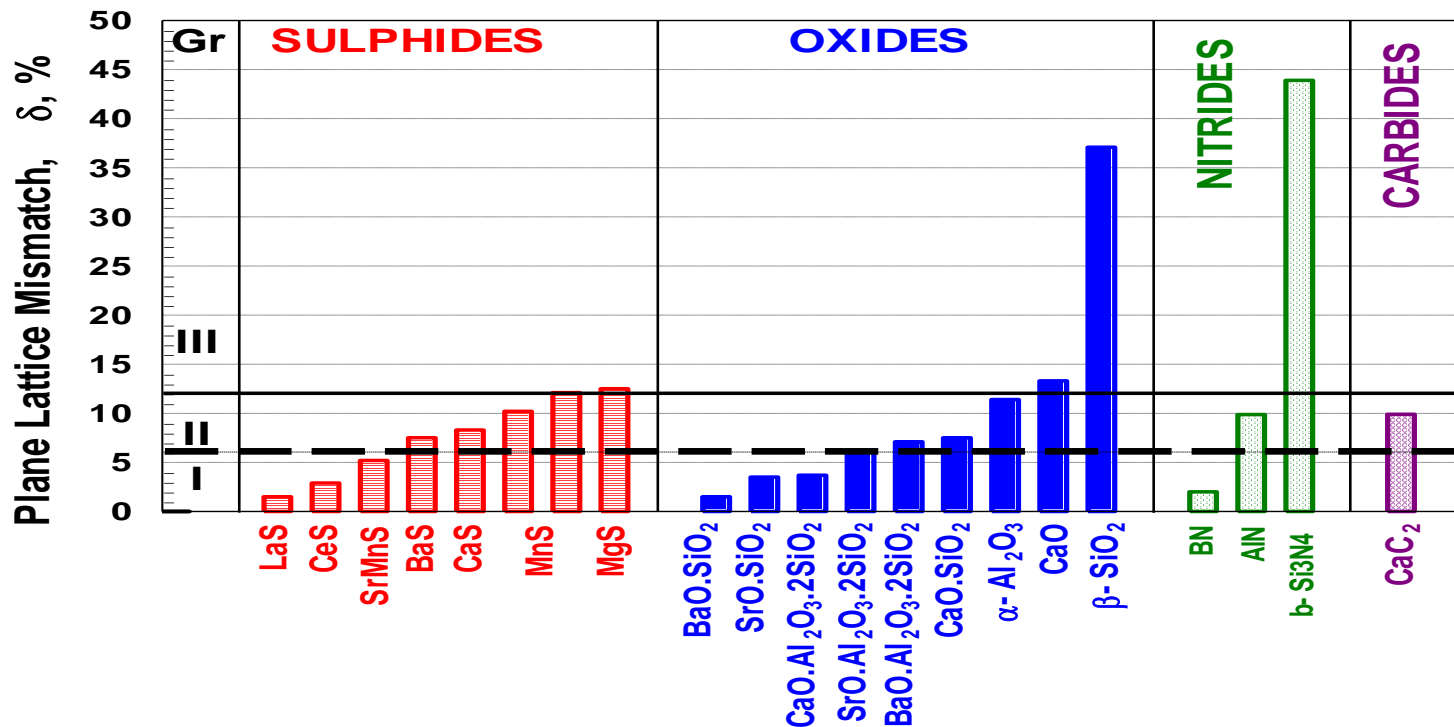
- ❖ **Examines how superheating affects the iron quality** *and*
 - which metallurgical treatments are effective.

I. HETEROGENEOUS GRAPHITE NUCLEATION IN COMMERCIAL GREY CAST IRONS

- **Only when liquid iron undercools sufficiently (200 - 230°C),**
 - can the smaller sizes of micro cluster (C₆)_n exist
 - as stable homogeneous nuclei for graphite particles.
- **Under normal cond., such high undercooling is difficult to achieve**
 - therefore, the nucleation of graphite is mainly by heterogeneous nucleation:
 - residual graphite particles *and / or* non-metallic microinclusions
- **High Residual Graphite dissolution in the base iron, *especially for***
 - Low-S (<0.05%S), High superheating (>1500°C) irons, *typically for*
 - electric melting, new generation medium frequency induction furnaces
- **Practically, only existed *or / and* newly formed micro-inclusions act as graphite nuclei in commercial grey irons, *depending on***
 - melting practice, iron chemistry, molten iron treatments, solidification
 - thermodynamic conditions of compounds formation, *and*
 - lattice parameters difference compound - graphite [δ , %]

Mismatch (δ , %) between a special lattice face of Nucleants and (0001) face of Graphite

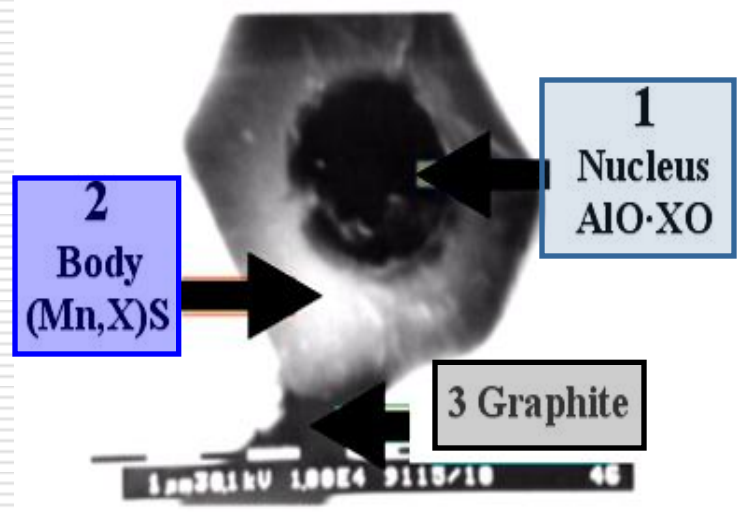
- I - $\delta < 6\%$ has the strongest nucleation ability
- II - $\delta = 6 - 12\%$ has nucleation ability
- III - $\delta > 12\%$ has a weak nucleation ability



[I. Riposan – 115th AFS Casting Congress, Schaumburg, USA, 2011, Panel Paper 11-131]

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A three-stage model for the nucleation of graphite in grey iron is proposed



1) small oxide-based sites [$< 3\mu\text{m}$] are early formed in the melt as stable microinclusions of deoxid. elements such as **Mn, Si, Al, Ti, Zr...**

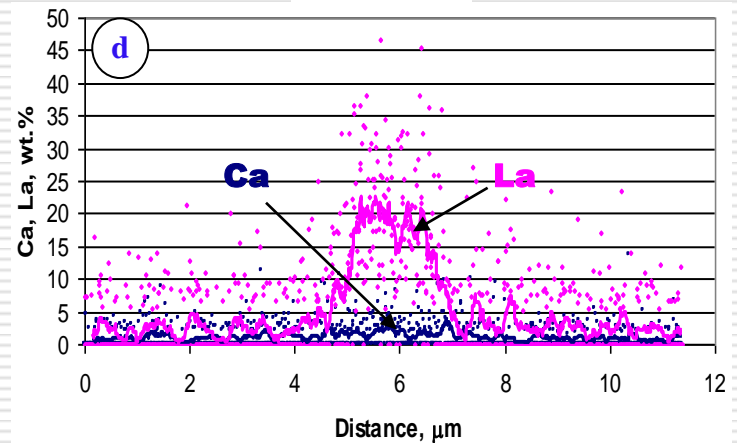
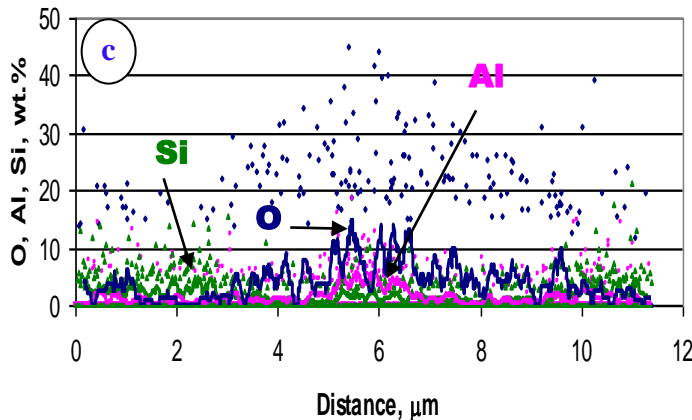
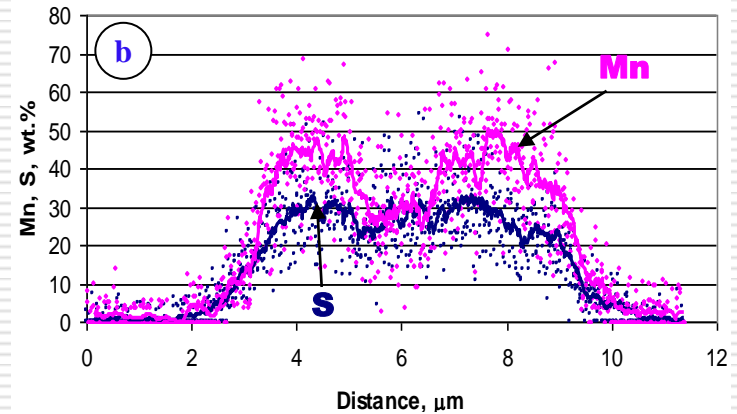
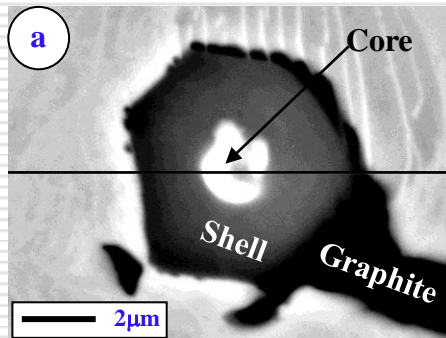
2) complex (Mn,X)S compounds [$1...10\mu\text{m}$] nucleate at these compounds [**Mn, S** importance] [**X = Fe, Si, Al, Zr, Ti, Ca, Sr, P, Ce, La...**]

3) graphite nucleates on the sides of the (Mn,X)S compounds with suitable crystallographic fit with graphite [**X-Inoculating Element** contribution]

[I. Riposan, M. Chisamera, S. Stan, T. Skaland. *AFS Trans.*, 2001 / *SPC17*, 2002 / *Int. J. Cast Met. Res.* 2003 / *AFS Inoc. Conf.* 2005 / C. Loper Symp., 2009 / *Mat. Sci. & Techn.*, 2010]

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Typical example of chemical analysis along cross-section of complex (Mn,X)S compound associated with graphite [a - Compo; b - Mn, S; c - O, Al, Si; d - Ca, La]



[I. Riposan, M. Chisamera, S. Stan, E. Stefan, C. Hartung – SPCI9-Luxor, Egypt, 2010;
Key Eng. Mater., 2011, V. 457, pp. 19-24.]

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II. LOW S / AI GREY CAST IRONS CHARACTERISTICS

Typical problems of low S, low Al grey cast irons

- Solidification with high eutectic undercooling

- * Carbides and / or Undercooled Graphite (type D)

- * Low number of Eutectic Cells

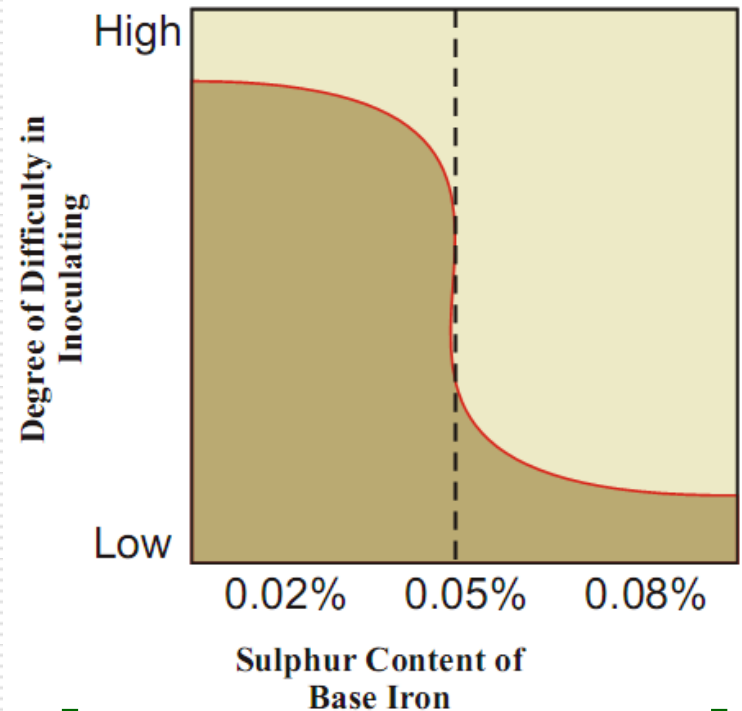
- * Non homogeneous structure

Limited response to inoculation----->

- Unfavourable conditions for type A Graphite formation

Inoculants: FeSiAlX: X = Ca, Ba, Sr....
traditionally inoculating elements

- to promote compounds [μm scale] in the iron melt, to act as active graphite nucleation sites, at lower eutectic undercooling

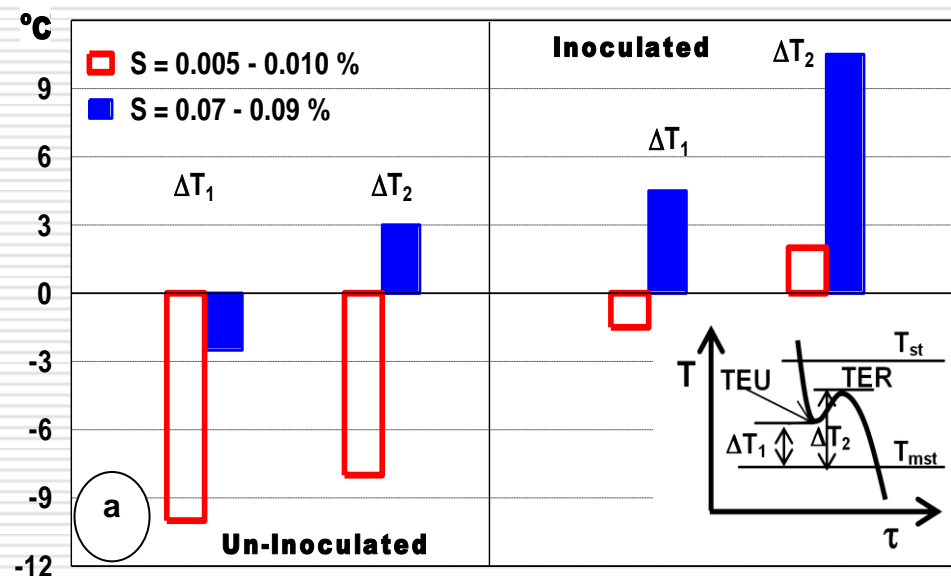


[www.foundry.elkem.com]

[I. Riposan, M. Chisamera, S. Stan G. Grasmø, C. Hartung, D. White.

113th AFS Casting Congress, 2009, Las Vegas, USA]

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Base Iron:
As-Melted vs
Resulphurized Irons
(%Mn) x (%S):
0.018 vs 0.055

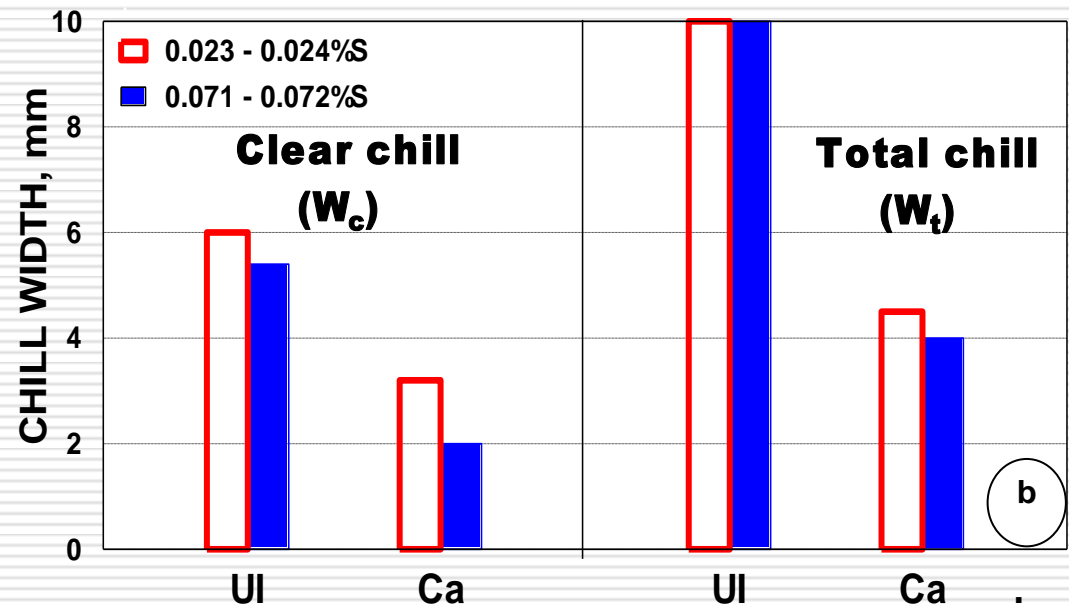
[M.Chisamera, I.Riposan, S.Stan,
D.White, G.Grasmo – *Asian Foundry*
Congress, 2008, Japan; Int. J. Cast Met.
Res., 2008, 21(1- 4), 39..]

Base Iron:
Ductile Iron vs
Grey Iron

$$[\Delta T_1 = TEU - T_{mst};$$

$$\Delta T_2 = TER - T_{mst}]$$

[M. Chisamera, I. Riposan, S. Stan, T.
Skaland. *64th World Foundry Congr.,*
2000, Paris, France]

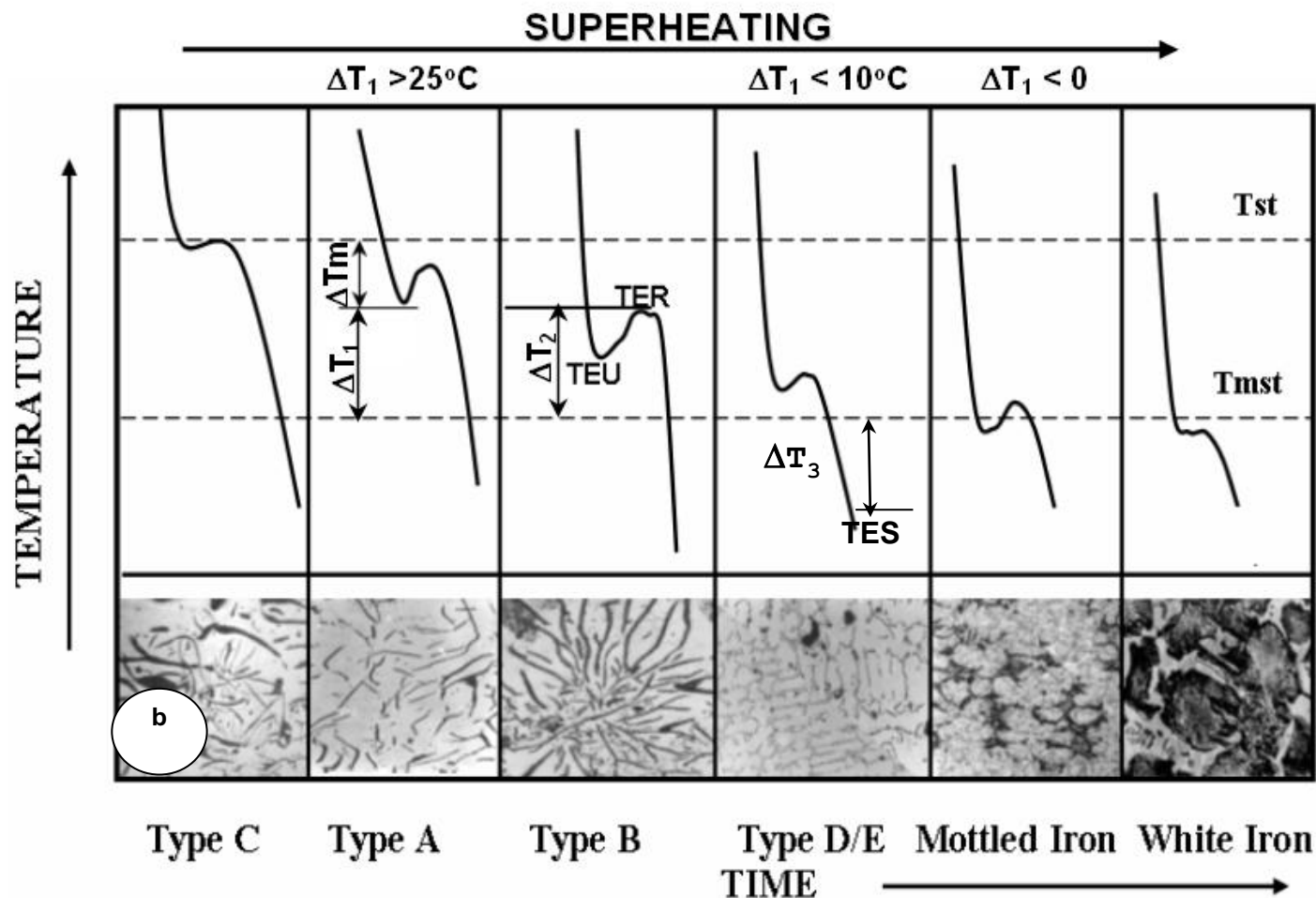


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III. SUPERHEATING OF IRON MELT IN ACID LINED CORELESS INDUCTION FURNACE [CIF]

- **There are different reasons for superheating grey iron in the acid lined, Coreless Induction Furnaces [CIF]**
 - (1) the production of thin wall (< 5 mm wall thickness) grey iron castings,
 - (2) correction after melting a steel charge needs power for stirring,
 - (3) post - furnace metallurgical treatment,
 - (4) numerous transfers and long holding time.
 - **The new generation of CIF: high superheat temp. [$> 1550^{\circ}\text{C}$], *due to***
 - high perform. electrical equipment with high specific power rating $> 250 \text{ kW/t}$
 - quality improvem. allow the acid refractory lining to withstand high temp.
 - high purity SiO_2 mater., optimum crystalline structure, **active agents addition**
 - **One application uses a SiC addition in a $\text{SiO}_2 - \text{B}_2\text{O}_3$ mass**
 - leading to improved lining life (20-100%),
 - especially with high rust levels in the metallic charge
 - inclusively added to sustain Mn oxidation for ferritic ductile iron
- [I. Riposan, M. Chisamera et al. Patent RO 110470-B1/1996; RO. Foundry J. 1996, 3, pp. 18-22]**

Superheating - Cooling Curves - Structure Relationship



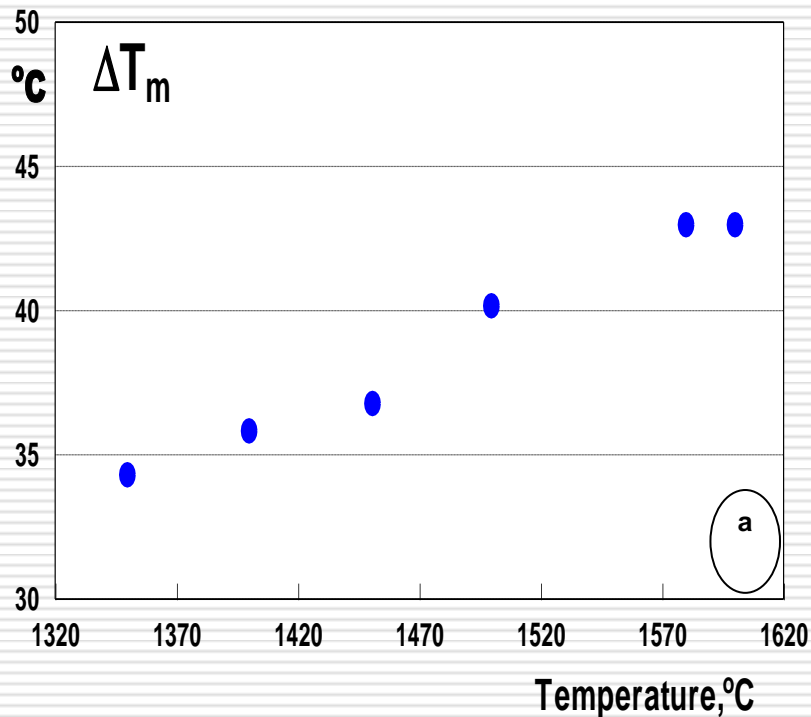
[S. Stan, I. Riposan, M. Chisamera, M. Barstow – Adv. Mater. Res., V. 23 (2007), pp. 307-310]

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Eutectic Undercooling and Chill Tendency in Superheated Grey Irons

[3.9%CE, < 0.02%S, (%Mn) x (%S) < 0.01, < 0.005%Al, < 0.0005%Zr]

Eutectic Undercooling [$\Delta T_m = T_{st} - TEU$]

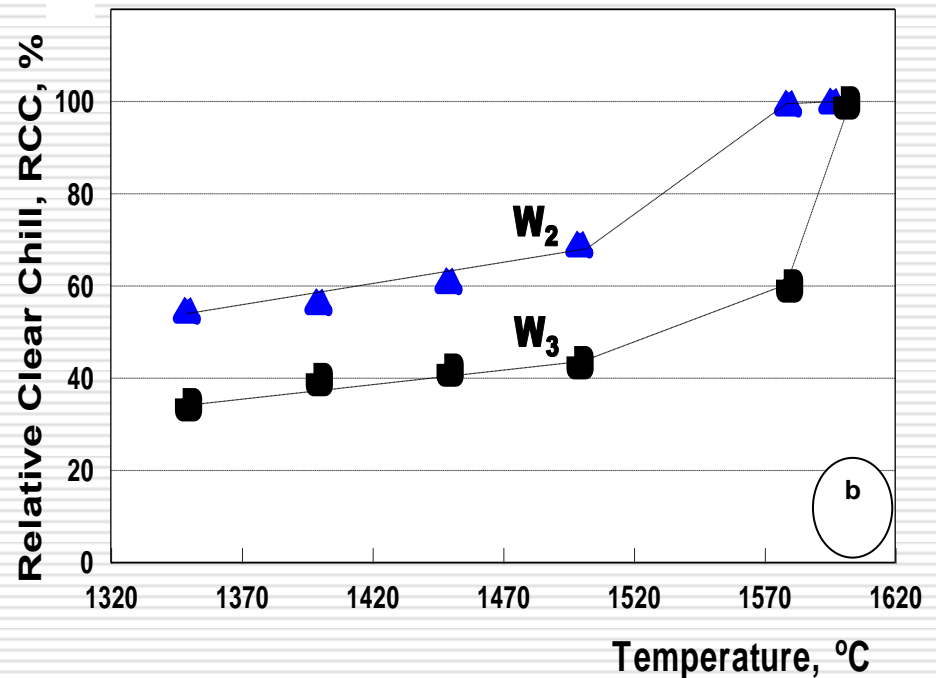


Chill Tendency

$$[\Delta T_m = T_{st} - TEU]$$

[W2, W3 – wedges ASTM A367]

$$[RCC = (W_c / B) \times 100]$$



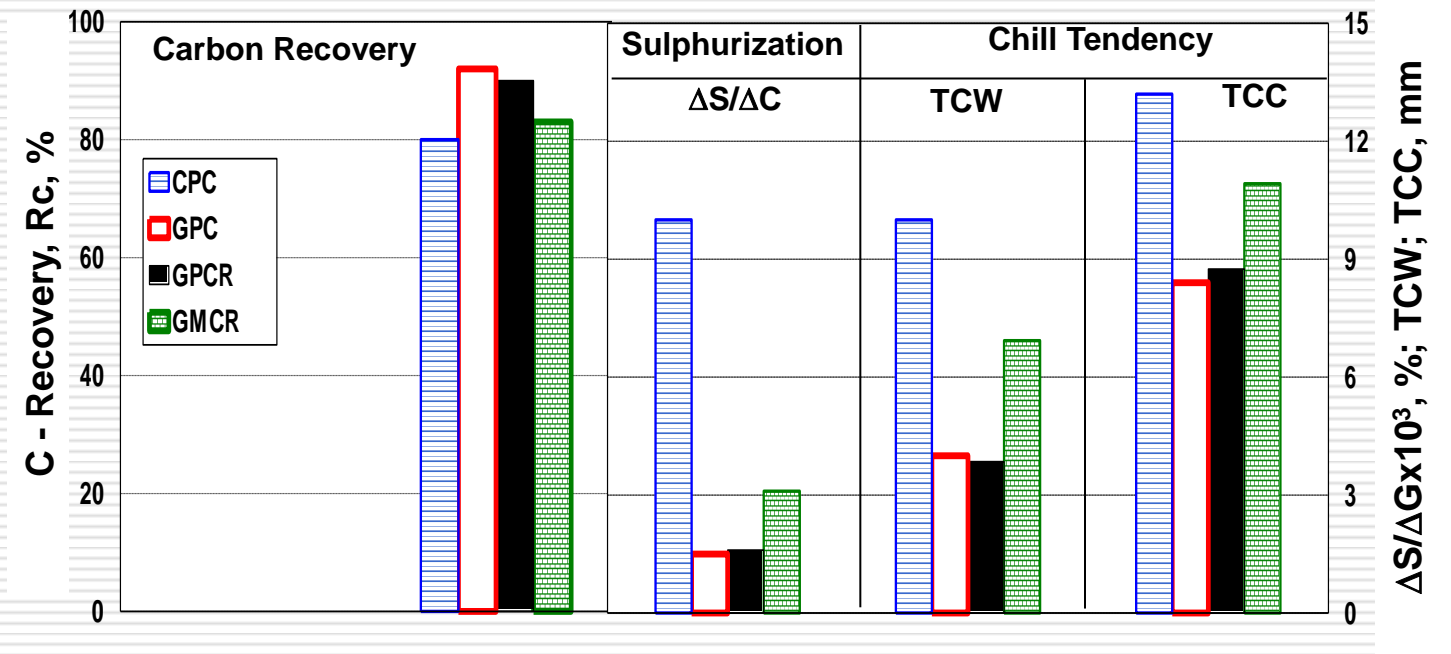
IV. METALLURGICAL TREATMENT OF ELECTRIC MELT GREY IRONS

A. PRE-TREATMENT OF THE IRON MELT IN A MELTING FURNACE

(1) Carbon materials

[Melt Iron: 3.0-3.25%CE, 2.5-2.9%C, 1.6-1.85%Si, 0.016-0.022%S, 1500°C]

[**CPC** – Calcined PC; **GPC** – Graphitized PC; **GPCR** – Graphitized Packing PC, a by-product; **GMCR** - Graphitized Packing Metallurgical Coke, a by-product]



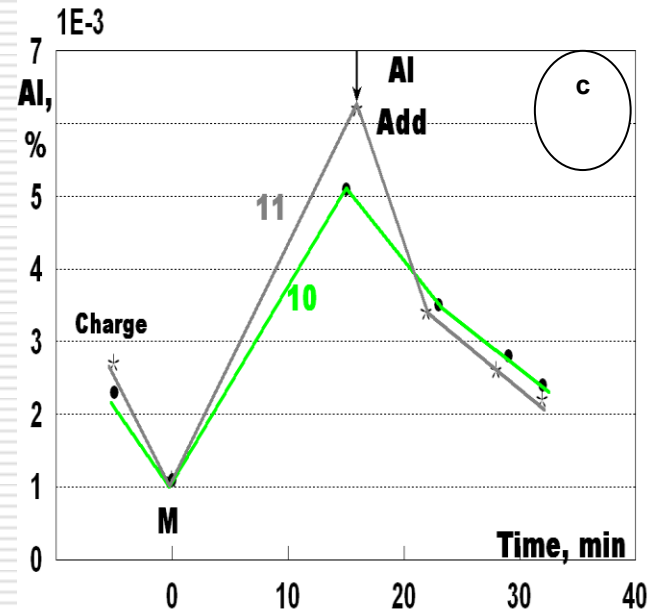
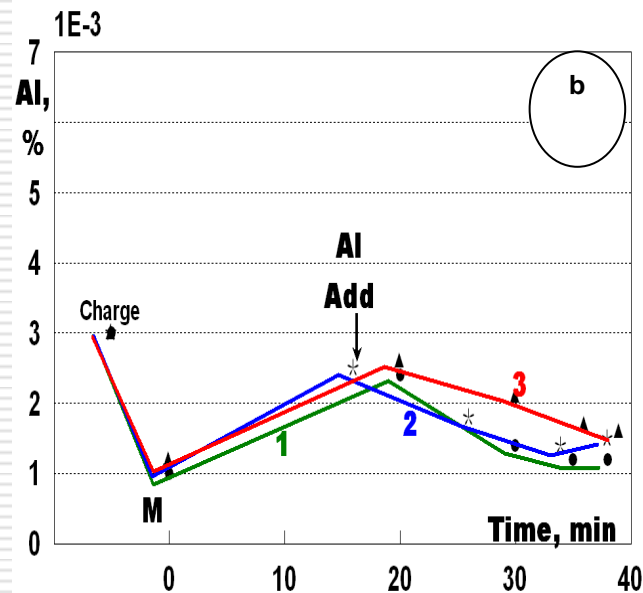
[M. Chisamera, I. Riposan, L. Stan et al. *67th World Foundry Congr.*, 2006, Harrogate, UK; *Adv. Mater. Res.* 2007, 23, 287]

(2) Metallurgical Silicon Carbide [Met. SiC]

- **Worldwide foundry experience reports benefits from using Met. SiC:**
 - reduced chill; improved machinability; improved and more uniform mech. properties;
 - less scrap: hard edges, shrinkage and gas porosity, graphite degeneration; dross.
 - increased Mg-yield; higher nodule count.
 - higher durability of furnace acid lining;
 - **The preconditioning / inoculation effect of Met. SiC, *promoted by***
 - presence of SiC molecules (not from melting but after dissolution), *and*
 - residual crystalline carbon, as graphite.
 - **This effect benefits the formation of:**
 - fine iron structure / lower chill (carbide forming)
 - reduction, or even elimination, of the inoculant addition.
 - **Possibility to improve the carburizing / preconditioning / inoculation performance of Met. SiC, *by***
 - optimizing the chemical composition,
 - especially by increasing the free graphite content
 - promotion of type – A graphite, less carbides
- [I. Riposan, M. Chisamera, I. Petrus, V. Pintea. *Patents* RO 105.063-A/1991, RO 101.922-A/1990 / 60th World Foundry Congr., 1993, Netherlands; ARABCAST Congress, 2004, Egypt]**

(c) Al recovery by charge and after melt. with in-furnace additions [MF – CIF, acid lined, 1500 – 1600°C]

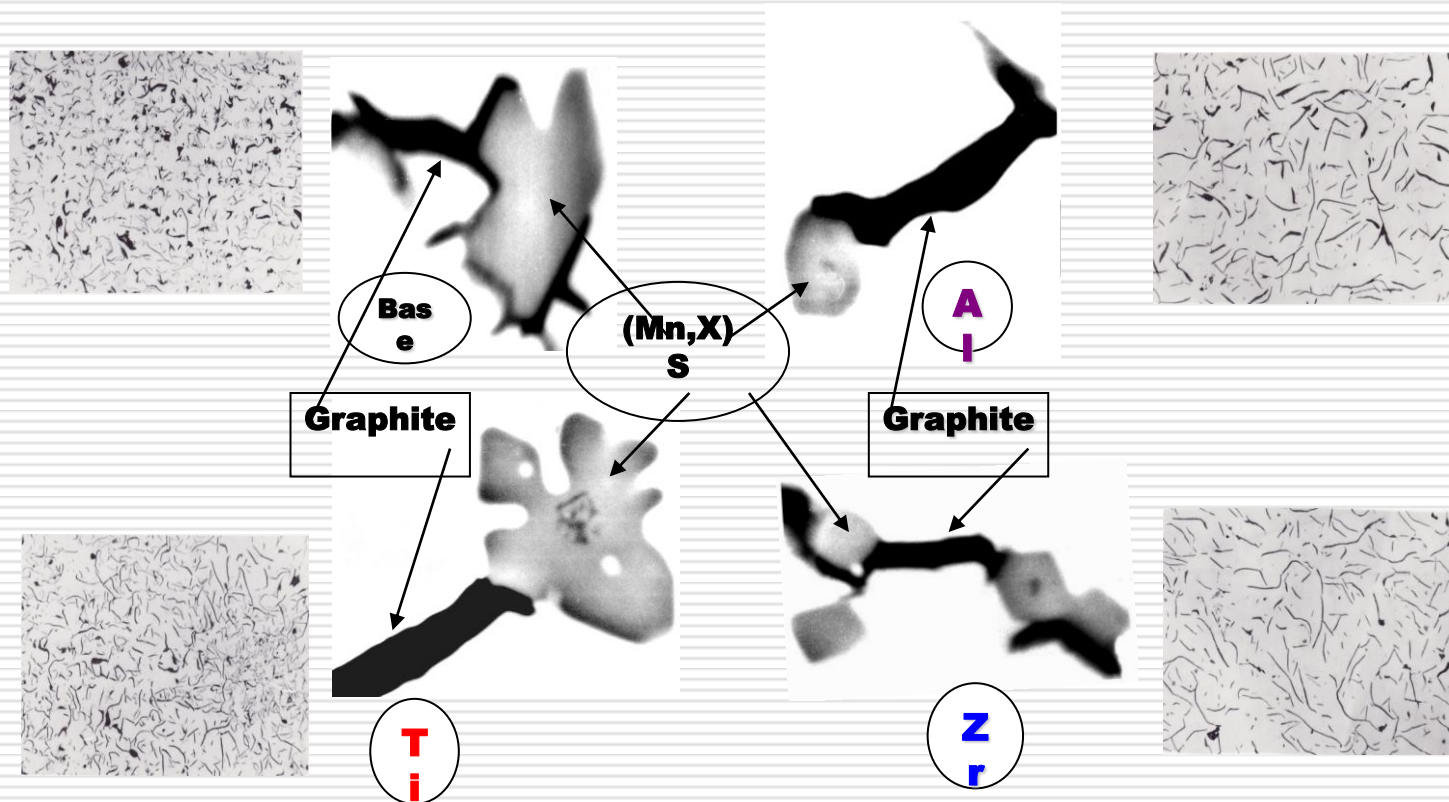
- **Al - a key element within complex (Mn,X)S acting as graph. nucl. sites**
 - experiments to control **Al** in acid lined, CIF melting with superheating
 - independent of the added **Al**, final **Al** was less than **0.003wt.% Al**
 - the final **Al** would be too low to sustain graphite nucleation
[> 0.004wt.%Al necessary]



[I. Riposan, M. Chisamera, S. Stan, G. Grasmø, C. Hartung, D. White. *AFS Trans.*, 2009, 117,423-434]

B. COMPLEX TREATMENTS OF ELECTRIC MELT GREY IRONS

(1) Al, Zr and Ti Preconditioning Comparison
[0.003 – 0.03%Al, 0.0005 – 0.015%Zr, 0.005 – 0.03%Ti]



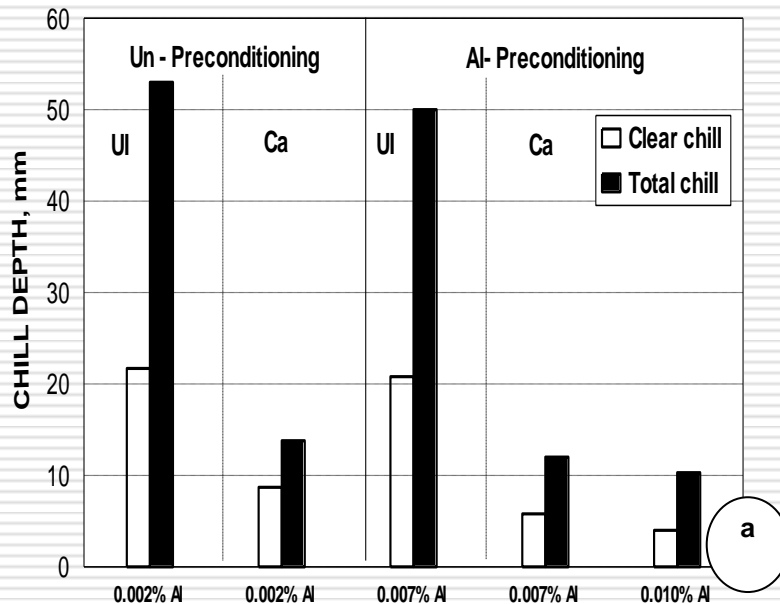
[I. Riposan, M. Chisamera, S. Stan, C. Hartung, D. White - *The Carl Loper Cast Iron Symp.*, 2009, Madison, USA / *Mater. Sci. Techn.*, 2010, 26 (12), 1439-1447]

(2) Double Treatments: Preconditioning + Inoculation

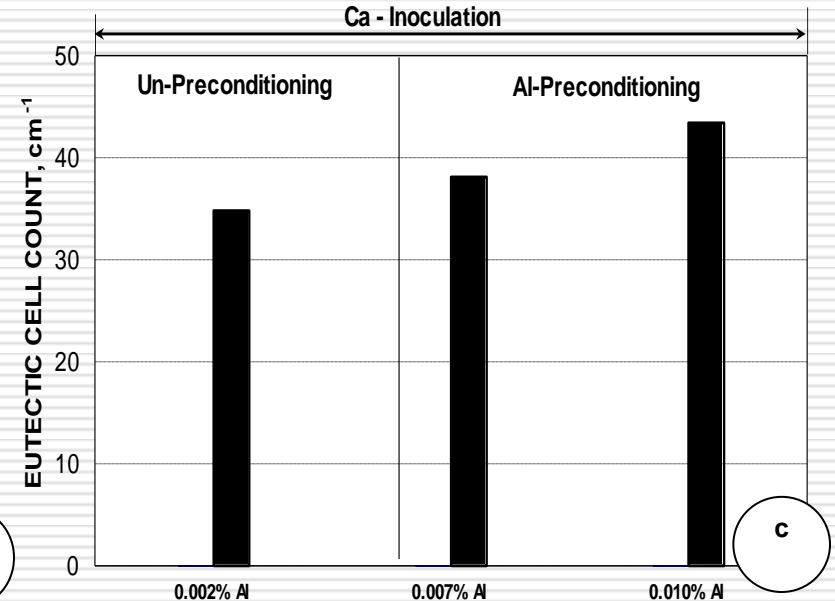
(a) Al preconditioning + Ca - FeSi inoculation

**[3.6-3.7%CE, 0.44-0.46%Mn, 0.09-0.1%S, (%Mn) x (%S) = 0.04-0.05,
0.002 / 0.007 - 0.01%Al, 0.0005%Zr, 1520°C]**

CHILL DEPTH [wedge sample]



EUTECTIC CELL COUNT

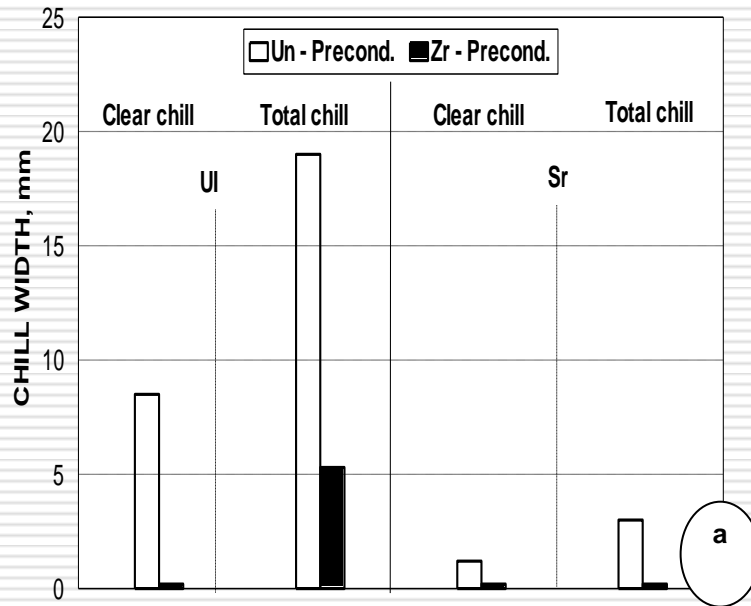


[M.Chisamera, I.Riposan, S.Stan, D.White, G.Grasmo – Asian Foundry Congr., 2008, Nagoya, Japan; Int. J. Cast Met. Res., 2008, 21(1-4), 39-44]

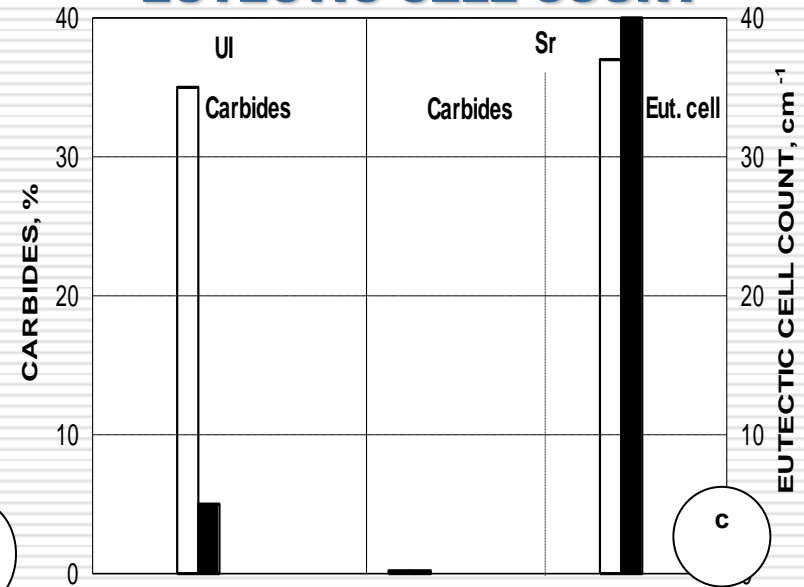
b) Zr preconditioning + Sr - FeSi inoculation

**[3.7-3.9%CE, 0.55-0.65%Mn, 0.095-0.1%S, (%Mn) x (%S) = 0.05-0.06,
0.002-0.004%Al, 0.0005 / 0.013%Zr, 1520°C]**

CHILL DEPTH [wedge sample]



CARBIDES / EUTECTIC CELL COUNT



[M.Chisamera, I.Riposan, S.Stan, D.White, G.Grasmo – Asian Foundry Congr., 2008, Nagoya, Japan; Int. J. Cast Met. Res., 2008, 21(1- 4), 39-44]

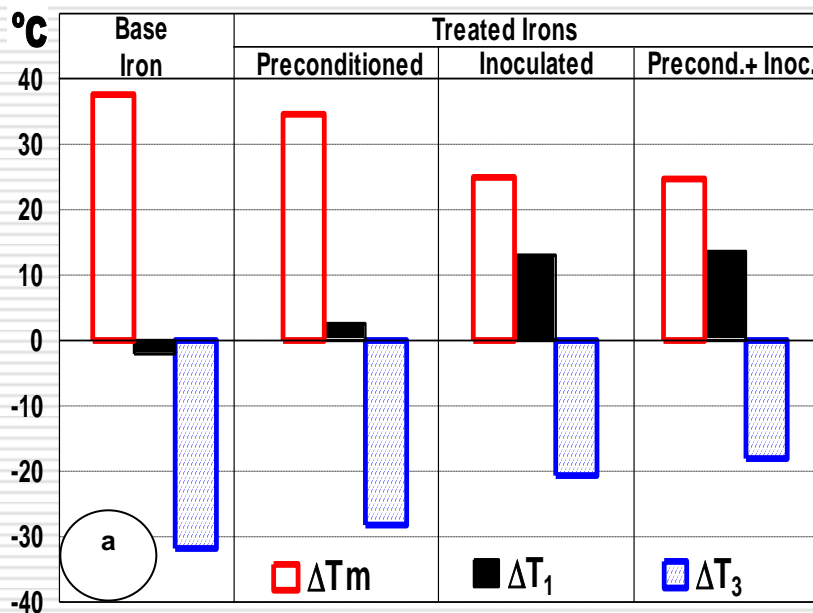
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c) Al,Zr,Ca - FeSi preconditioning + Ca,Ba - FeSi inoculation

**[3.9-4.0%CE, 0.7-0.8%Mn, 0.012-0.015%S, (%Mn) x (%S) < 0.015,
< 0.005%Al, < 0.0005%Zr, 1530°C]**

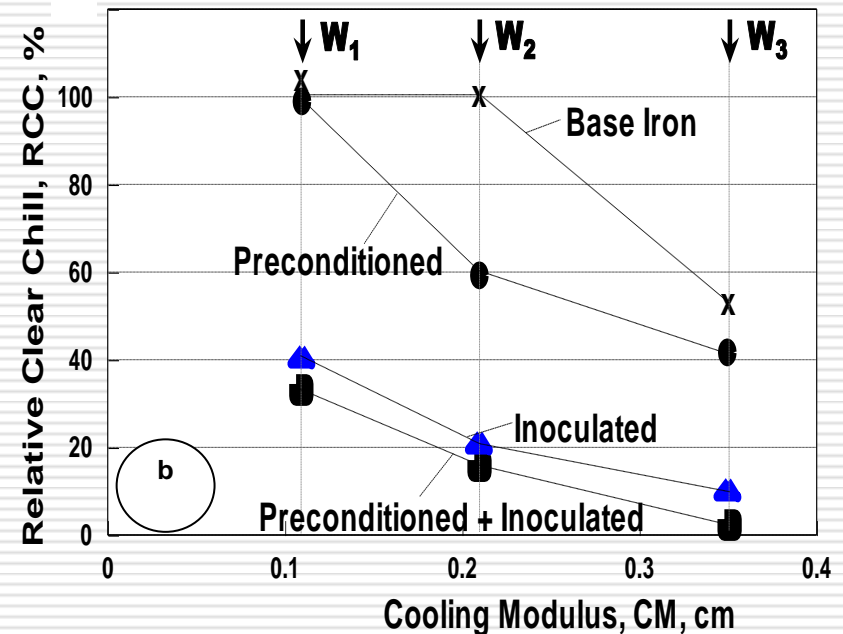
EUTECTIC UNDERCOOLING

**$\Delta T_m = T_{st} - TEU$; $\Delta T_1 = TEU - T_{mst}$;
 $\Delta T_3 = TES - T_{mst}$**



CHILL TENDENCY

**W_1, W_2, W_3 – wedges ASTM A367]
 $[RCC = (W_c / B) \times 100]$**



RECOMMENDATIONS FOR FOUNDRY APPLICATION

OBJECTIVE

To sustain (Mn,X)S Compounds formation, compatible to graphite nucleation at lower eutectic undercooling:

- lower size / higher compactness / higher count particles suitable crystallographic fit with graphite

EXPECTED FINAL RESULTS

- ✓ **Solidification at lower eutectic undercooling / recalescence**
- ✓ **Graphite nucleation at lower undercooling**
 - **A-type graphite promotion / D-type graphite avoidance**
- ✓ **Carbides avoidance**
- ✓ **Higher eutectic cell count**
- ✓ **Thin wall grey iron castings especially**
 - **no Carbides or / and D - type graphite**

Three groups of elements are important to sustain graphite nucleation in grey irons

- 1) strong deoxidizing elements**, such as **Al** and **Zr**, to promote formation of very small micro-inclusions [**0.005 – 0.010%Al** benef.; **I. Riposan et al**]
- 2) Mn** and **S** to sustain **MnS** type sulphide formation
[(%**Mn**) x (%**S**) = **0.03 – 0.06**; **R. Gundlach**]
- 3) inoculating elements** (such as **Ca, Sr** etc) which act in the first stage and / or in the second stage of graphite formation, to improve the capability of **(Mn,X)S** compounds to nucleate graphite.

Three technological steps appear to be necessary to produce high performance grey cast iron

- 1) furnace super-heating** of the molten iron, to eliminate un-desired heredity influence of charge:
 - depending on the charge characteristics *and* melting practice
- 2) pre-conditioning** with oxide forming elements (**Al** *or / and* **Zr**) of the base iron, especially for:
 - low quality charge materials *or / and* melting practice, *or*
 - low performance inoculation procedure *or / and* pouring practice
- 3) inoculation** (such as **Ca, Sr....**) of pre-treated iron, as final treatment:
 - to control eutectic undercooling
 - *according to the quality of base iron
 - *according to the castings characteristics

FOUNDRIY APPLICATION RECOMMENDATIONS

No.	Step	Characteristics
1	Melting	<ul style="list-style-type: none">* Minimum 1500°C superheat / Induction furnace* Scrap steel and Iron charge;* C and Si correction;* Met. SiC beneficial;* Recovered (by-product) graphit. carbon mater. beneficial;* S correction for optimum (%Mn) x (%S) Factor [0.03-0.06] - generally, in a 0.4-1.2%Mn / 0.04-0.12%S range
2.	Precon- ditioning	<ul style="list-style-type: none">* Al, Zr - FeSi alloys beneficial* 0.05 – 0.15wt. % addition before furnace tap (0.1 % usually)* 0.005-0.015wt.%Al / Zr in final iron [max. 0.01 % usually]
3.	Inocula- tion	<ul style="list-style-type: none">* Different inoculation techniques* Normal to lower addition rates* Sr - FeSi / Ca - FeSi / Ca,Zr - FeSi / Ca,Ba - FeSi alloys

ACKNOWLEDGMENTS

❖ **Laboratory research and plant trials were conducted by POLITEHNICA University of Bucharest, Romania.**

❖ **Authors thank to ELKEM AS Foundry Products Division, Norway, which partially supported the laboratory experiments and approved reference to those results for publication.**

THANK YOU

The image features the words "THANK YOU" in a large, 3D, sans-serif font. Each letter is filled with a vibrant rainbow gradient, transitioning from purple at the top to red, orange, yellow, green, and blue at the bottom. The letters are outlined with a thin white border and a dotted red line. They are positioned on a white surface, casting soft, grey shadows to the left and slightly forward. The background is a solid, light grey.